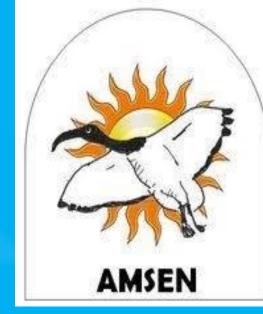


# Spectroscopy of trivalent neodymium ions (Nd<sup>3+</sup>) in zinc oxide (ZnO) powders and thin films

N.C. Gatsi<sup>1,2,3</sup>, M. Mujaji<sup>1,3</sup> and D. Wamwangi<sup>1,3</sup>





<sup>1</sup> School of Physics, University of the Witwatersrand, Johannesburg, <sup>2</sup>African Materials Science and Engineering Network (A Carnegie-IAS RISE Network), <sup>3</sup>DST-NRF Centre of Excellence in Strong Materials, hosted by the University of the Witwatersrand.

## 1. INTRODUCTION AND MOTIVATION

#### Trivalent rare-earth ions (RE<sup>3+</sup>) doped ZnO

- ZnO is a wide-band-gap semiconductor with a wurtzite structure with potential for RE incorporation.
- Intrinsic point defects (vacancies, antisites, interstitials) in ZnO are responsible for its broad emission band(s) in the VIS-NIR region.
- RE<sup>3+</sup> ions doped into ZnO yield narrow and discrete absorption and emission peaks superposed on the broad ZnO emission band.
- RE<sup>3+</sup> ions emission finds application in bio-imaging, lasers and colored-light production while absorption is useful on welder's goggles for blocking the yellow glow.
- Most of the reported ZnO:Nd3+ emission is for UV excitation and at room temperature. [1]

### Research Aim

 Spectroscopic study of Nd<sup>3+</sup> ions in ZnO powders and thin films using VIS excitation and at sample temperature of 10 K.

#### 2. BACKGROUND

- The energy levels of intrinsic defects in ZnO lie within its energy gap (Figure 1).
- Some energy levels of the Nd<sup>3+</sup> ions lie within the ZnO energy gap (0-27 700 cm<sup>-1</sup> (361 nm)) .[1]
- Indirect excitation of the Nd multiplet within the ZnO gap is possible through excitation of the intrinsic defects and subsequent energy transfer to the Nd<sup>3+</sup> ions.
- Co-doping with lithium (Li) enhances the Nd emission.
- Laser spectroscopy is a non-destructive method for studying the photoluminescence properties of rare-earth ions doped into ZnO.<sup>[2]</sup>

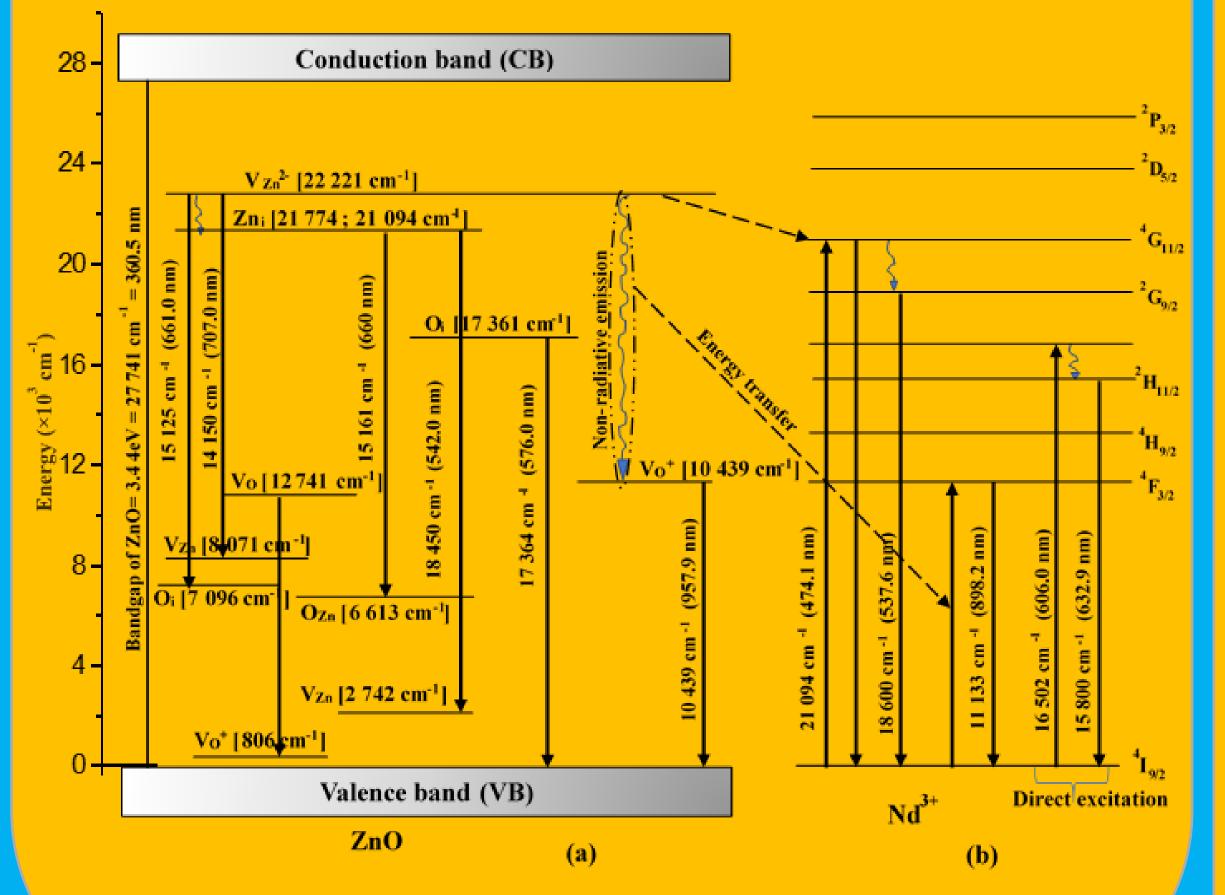


Figure 1: (a) Some defects states which lie within the bandgap of ZnO and (b) energy states of Nd<sup>3+</sup> ions.

## 3. EXPERIMENTAL

## a. Samples preparation

- Powder samples were prepared by the sintering technique (grinding, compaction by pressing and annealing).
- Thin films were prepared by radio-frequency magnetron sputtering technique using a 3mol%Nd doped ZnO target.
- Samples (both powders and films) were annealed at varying temperatures (600–950°C), in air and for 2 hrs. A ramping rate of 5°C per/min was adopted.

## b. Structural characterization

 Surface morphology and elemental composition was investigated by SEM and EDS, respectively.

## c. Spectroscopy

- Four Ar+ laser lines (457.9 nm, 476.5 nm, 488.0 nm and 514.5 nm) were used for excitation.
- Samples were cooled to 10 K in a closed-cycle cryostat.
- resolution Emission was detected by a high monochromator and a PMT.

## 4. RESULTS AND ANALYSIS

Structural characterization of the powders

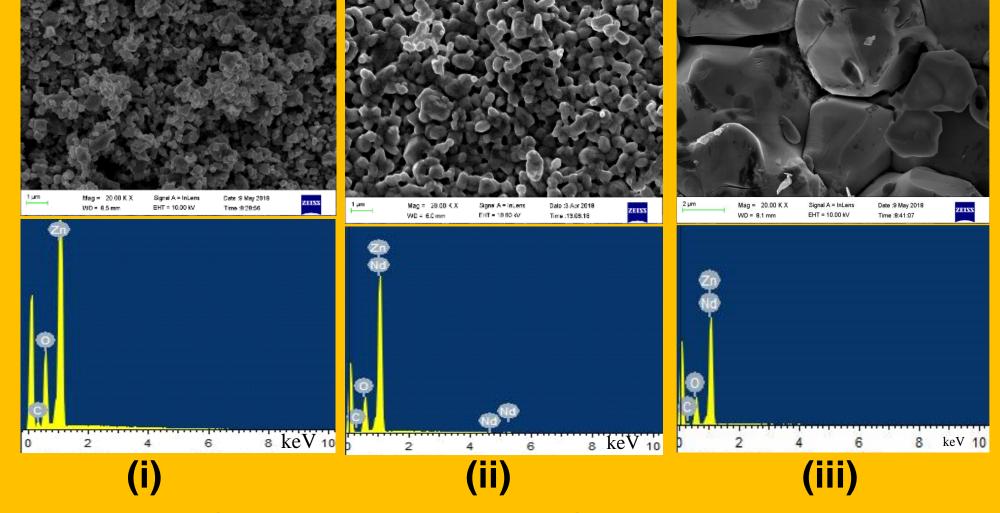


Figure 2: SEM images and EDS spectra of:

- i. ZnO:1mol%Nd annealed at 600°C,
- ii. ZnO:1mol%Nd annealed at 950°C
- iii. ZnO:1mol%Nd:10mol%Li annealed at 950°C
- The combined effect of Li co-doping and annealing at 950°C is larger grains (1000%) and aggregation of Nd<sup>3+</sup> ions at the grain boundaries.
- From EDS spectra, the average surface concentration of Nd is 0.6 at.%.

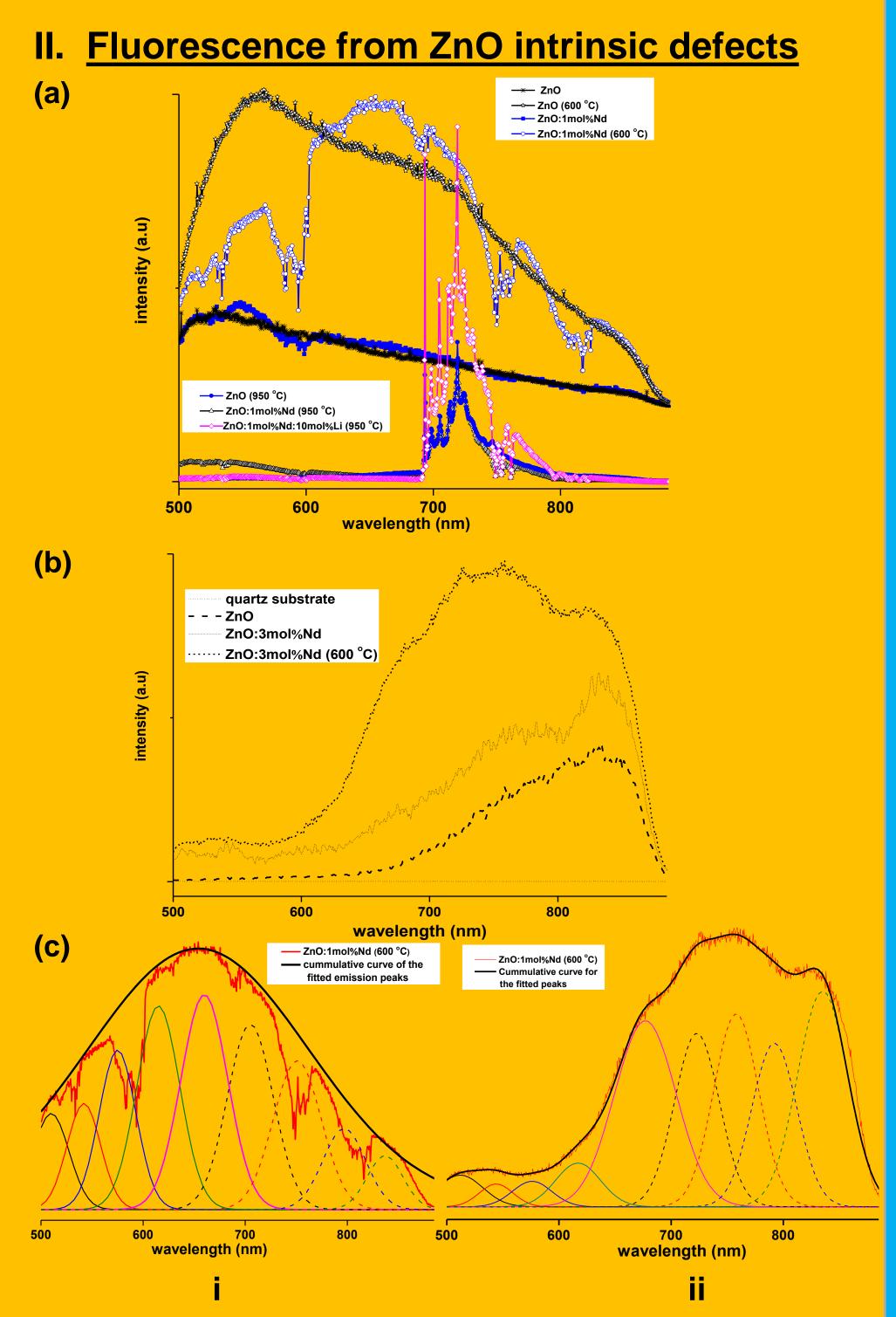


Figure 3: Effects of doping and annealing on fluorescence from (a) powders and (b) films. The deconvoluted intrinsic-defect peaks are shown in (c) i and (c) ii for the powders and thin films, respectively.

Table 1: Intrinsic defects emission peak positions, the corresponding transitions and the nearresonant Nd<sup>3+</sup> ion multiplet (Figure 1).

Powders		transitions	Nd multiplet(s)
l l	Films		
510	512	$Zn_i^+ \rightarrow O_i$ or $V_{Zn}$	<sup>2</sup> <b>G</b> <sub>9/2</sub>
542	544		3/2
576	576	$O_i \rightarrow VB;$	<sup>2</sup> <b>G</b> <sub>7/2</sub>
615	617	$Zn_i^+ \rightarrow V_{Zn}^-$	<sup>4</sup> G <sub>5/2</sub>
660	677	$Zn_i^+ \rightarrow O_{Zn}$ or $O_i$	<sup>2</sup> H <sub>11/2</sub>
705	723	$\begin{array}{c} V_{Zn}^{2\text{-}} \rightarrow O_i \ or \ V_{Zn}, \\ V_O \rightarrow VB \ , \ V_{Zn}^{2\text{-}} \rightarrow V_{Zn} \ , \\ Zn_i \rightarrow \ V_{Zn} \end{array}$	<sup>4</sup> F <sub>9/2</sub>
751	758	$V_{O} \rightarrow V_{O}^{+} or V_{Zn}$ ,	<sup>4</sup> S <sub>3/2</sub> ; <sup>4</sup> F <sub>7/2</sub>
797	792		512 7 11Z
837	835	$V_{Zn}^{2-} \rightarrow O_i$	<sup>2</sup> H <sub>9/2</sub> ; <sup>4</sup> F <sub>5/2</sub>

Both the powders and the films gave the same emissions attributed to intrinsic defects.

- The 457.9 nm excitation wavelength gave much higher emission intensities than other Ar+ laser lines.
- Absorption features in the broad ZnO emission profile indicate the Nd<sup>3+</sup> ions absorbed near-resonant energy emitted by the ZnO intrinsic defects.
- Annealing causes re-distribution of intrinsic defects as shown by the changed emission intensity distribution (Figure 3).

#### III. Fluorescence:Nd<sup>3+</sup> extrinsic defects

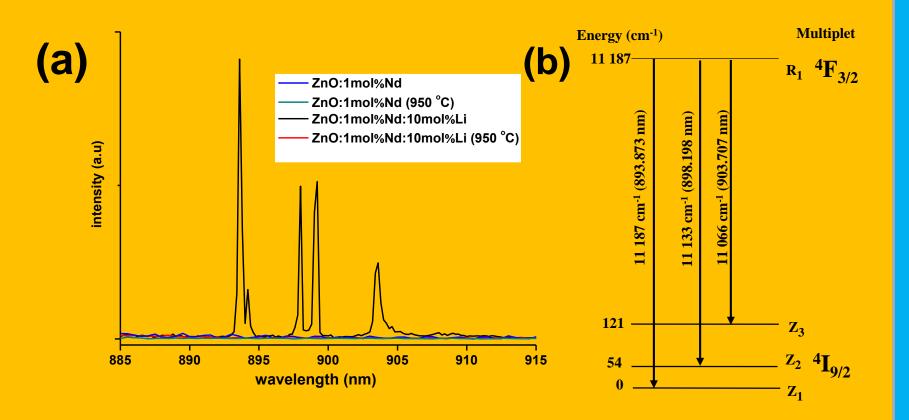


Figure 4: (a) Emission spectrum and (b) energy levels for ZnO:1mol%Nd:10mol%Li powder annealed at 950°C.

 Co-doping with Li<sup>+</sup> ions and annealing at 950°C promotes the population of V<sub>0</sub>+ defects which are linked to the population of the <sup>4</sup>F<sub>3/2</sub> multiplet (Figure 1).

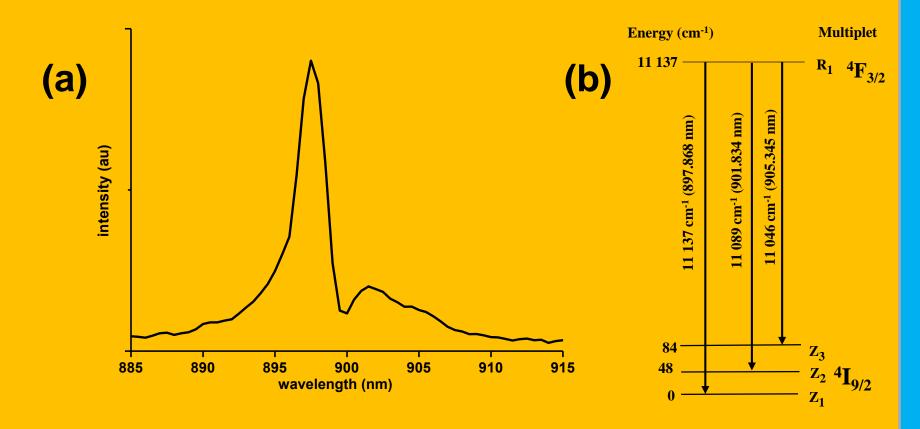


Figure 5: (a) Emission spectrum and (b) energy levels for ZnO:3mol%Nd thin films

## 5. CONCLUSION

- Nd<sup>3+</sup> Intrinsic defect emission absorption is observed in the 500-850 nm region.
- Nd<sup>3+</sup> emission is only observed near 900 nm  $(^4F_{3/2} \rightarrow ^4I_{9/2})$  in the Li co-doped samples annealed at 950°C and the films.
- The energy level positions of Nd<sup>3+</sup> ions are in the same region in the films and the powders; the broader film transitions could be an indication of the quality of the films.
- Doped sintered-powders can be costeffective targets for thin film deposition by the pulsed laser technique.
- XRD analysis will be done to assess effects of doping and annealing.

## 6. ACKNOWLEDGEMENTS

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## 7. REFERENCES

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